



# Structure dependence of Nb<sub>2</sub>O<sub>5-x</sub> supported manganese oxide for catalytic oxidation of propane: Enhanced oxidation activity for MnO<sub>x</sub> on a low surface area Nb<sub>2</sub>O<sub>5-x</sub>

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## ABSTRACT

A series of Nb<sub>2</sub>O<sub>5-x</sub> with different structures were prepared as a carrier to manganese oxide catalysts for total oxidation of propane. The results demonstrated that a monoclinic structure of Nb<sub>2</sub>O<sub>5-x</sub> pre-calcined at 1025 °C, leads to significantly and surprisingly higher catalytic oxidation activity when MnO<sub>x</sub> is deposited at 400 °C even with extremely low specific area (around 3.94 m<sup>2</sup>/g) relative to the performance of MnO<sub>x</sub>/Nb<sub>2</sub>O<sub>5-x</sub> (~ > 50 m<sup>2</sup>/g pre-calcined at 500 °C). Conversion vs temperature profiles for fresh and aged catalysts were generated and performance compared for different materials. Brunauer Emmett Teller (BET) and X-ray diffraction (XRD) were conducted to reveal the textural and structural features of niobium-based catalysts. Raman, X-ray photoelectron spectroscopy (XPS) and temperature programmed reduction (TPR) were performed to further understand the interaction between manganese and niobium oxides. Raman spectra indicated a new Nb-O-Mn species formed due to the strong interaction between the activated niobium oxide carrier at 1025 °C and manganese oxide. This study describes a synergistic catalytic oxidation effect between Mn oxides deposited on a specific phase structure of Nb<sub>2</sub>O<sub>5-x</sub> with very low specific surface area. The enhanced catalytic performance is directly related to the proper ratio of Mn<sup>3+</sup>/Mn<sup>4+</sup> coupled with high ratio of Nb<sup>4+</sup>/Nb<sup>5+</sup> on the surface and the oxygen vacancies generated between monoclinic Nb<sub>2</sub>O<sub>5-x</sub> and MnO<sub>x</sub>.

## 1. Introduction

It is well known that volatile organic compounds (VOCs) represent a large group of chemicals, toxic to human health due to their malodorous, mutagenic and/or carcinogenic nature, but also detrimental to air quality via their role in photochemical smog generation and ozone depletion [1–3]. VOC's emission sources include pharmaceutical and chemical plants, petroleum refineries, automobile manufacturing and transportation fuels, etc. [4]. Consequently, to improve the world's environment and meet the ever-increasing stringent environmental legislation their catalytic abatement has been an active commercial and research topic for many years [5–7].

Propane (C<sub>3</sub>H<sub>8</sub>) is being produced in great amounts due to extraction of shale gas and liquefied petroleum gas. In addition, the oxidation of propane (C<sub>3</sub>H<sub>8</sub>) could be considered a prototype pollutant representing volatile organic compounds [8,9]. Accordingly, catalytic materials have been investigated and developed using propane

oxidation as a model compound [5,8,10–12]. It has been reported that supported noble metals (Pt and Pd), perovskite-type catalysts (LaMnO<sub>3</sub>) and transition metal oxides (Mn, Co, Fe, Cu) are catalytic component for the total oxidation of VOCs [13–17]. However, noble metals are indeed the most active and stable materials but their limited availability and expense are disadvantages which have stimulated the research of alternative non-precious metal oxides. Perovskite-type catalysts show high thermal stability but complicated preparation methods are usually needed to achieve their desired functional structure [14,15]. In contrast, simple transition metal oxides attract interest and exhibit considerably activity and durability but are less active than noble metals [6,8,9].

It has been reported that manganese oxides could offer promising catalytic activity towards the oxidation of propane. For example, Mn<sub>3</sub>O<sub>4</sub> and Mn<sub>2</sub>O<sub>3</sub> are known for good catalytic activity for catalytic combustion of VOCs [12,13,16,17]. Manganese oxides are also good candidates for total catalytic oxidation of carbon monoxide and the

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activity among manganese species are reported in the order of  $\text{Mn}_2\text{O}_3 > \text{MnO}_2 > \text{MnO}$  [18]. Its activity is attributed to the high redox properties and high mobility of lattice oxygen, which can be strongly enhanced by the combination with other metals [13,16–19].

Various high surface area carriers for manganese oxides such as  $\gamma\text{-Al}_2\text{O}_3$ ,  $\text{TiO}_2$ ,  $\text{SiO}_2$  and  $\text{ZrO}_2$  have been routinely studied. Environmentally friendly niobium oxide has been drawing some attention since it is not only a reducible oxide with oxygen storage capacity [20–23], but also provides a strong metal-support ( $\text{Nb}_2\text{O}_5$ ) interaction (SMSI) with catalytic metal oxides altering the oxidation states of catalytic metals [24]. The catalytic properties of niobium compounds have been studied quite extensively [24–31], but none have been used commercially for pollution abatement applications.

A paper from our laboratory by Xiang et al. [30] demonstrated that  $\text{Nb}_2\text{O}_5$ , pre-calcined at 800 °C in air and used as a carrier for cobalt oxide showed remarkably high CO and  $\text{C}_3\text{H}_8$  oxidation activity and stability relative to cobalt oxides on high surface area  $\text{Nb}_2\text{O}_5$  or  $\text{Al}_2\text{O}_3$ . Surprisingly comparable behavior to  $\text{Pt}/\text{Al}_2\text{O}_3$  was also reported in the same paper.

This current paper is a more detailed and systematic study of the impact of crystal structure of  $\text{Nb}_2\text{O}_5$ , generated by pre-calcination, on the catalytic oxidation activity of supported  $\text{MnO}_x$ . It provides mechanistic insight into the structural interaction between Mn oxides and high temperature monoclinic  $\text{Nb}_2\text{O}_5$  for the catalytic oxidation of VOCs. All catalysts were characterized by BET, XRD,  $\text{H}_2$ -TPR and XPS, and the performance evaluated mainly in the catalytic oxidation of propane, chosen as model VOCs molecule. In addition, the results will be compared with the alumina-based catalysis using similar catalytic preparation and test conditions. The thermal stability of the optimized catalyst was also studied and substantiated.

## 2. Experimental

### 2.1. Catalyst preparation

Powdered Niobium hydrate ( $\text{Nb}_2\text{O}_5 \cdot 5\text{H}_2\text{O}$ ) CBMM, San Paolo, Brazil) (BAFS) and  $\gamma$ -Alumina ( $\text{Al}_2\text{O}_3$ ) (Sigma-Aldrich) were individually calcined in air from 500 °C to 1100 °C with a ramp rate of 3 °C/min to a the final temperature for 3 h. Upon cooling to room temperature various loadings of manganese nitrate hydrate ( $\text{Mn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ ) (Sigma-Aldrich) were impregnated. The niobium hydrate calcined at 500, 700, 800, 900 and 1025 °C were given the notation of  $\text{Nb}_2\text{O}_5$ -x@500,  $\text{Nb}_2\text{O}_5$ -x@700,  $\text{Nb}_2\text{O}_5$ -x@800,  $\text{Nb}_2\text{O}_5$ -x@900 and  $\text{Nb}_2\text{O}_5$ -x@1025, respectively. The pre-calcined alumina was labelled  $\text{Al}_2\text{O}_3$ @500,  $\text{Al}_2\text{O}_3$ @800 and  $\text{Al}_2\text{O}_3$ @1025.

A set of catalysts synthesized using incipient wetness impregnation and designated, Z%  $\text{MnO}_x$ /(1-Z%) $\text{Nb}_2\text{O}_5$ -x (Y), where Z = 4, 6, 10, 20, 30 wt%, and Y =  $\text{Nb}_2\text{O}_5$ -x pre-calcined at 500 °C, 700 °C, 800 °C, 900 °C and 1025 °C. Each finished catalyst was dried in air at 110 °C overnight and heated to 400 °C at a rate of 2 °C/min and held at 400 °C in air for 3 h. Thermal Gravimetric Analysis (TGA) in air was used to establish 400 °C as adequate for complete decomposition of the nitrates. Similar methods and preparation conditions were used for the alumina-based catalysts, Z%  $\text{MnO}_x$ /(1-Z%) $\text{Al}_2\text{O}_3$  (Y). Catalysts studied of greatest interest were, Fresh 20%  $\text{MnO}_x$ / $\text{Nb}_2\text{O}_5$ -x(500) @400 °C, Fresh 20%  $\text{MnO}_x$ / $\text{Nb}_2\text{O}_5$ -x(800)@400 °C, Fresh 20%  $\text{MnO}_x$ / $\text{Nb}_2\text{O}_5$ -x(1025) @400 °C, Fresh 20%  $\text{MnO}_x$ / $\text{Al}_2\text{O}_3$ (500)@400 °C, Fresh 20%  $\text{MnO}_x$ / $\text{Al}_2\text{O}_3$ (800)@400 °C, Fresh 20%  $\text{MnO}_x$ / $\text{Al}_2\text{O}_3$ (1025)@400 °C, labelled as M-N500, M-N 800, M-N 900, M-N 1025, M-A500, M-A800 and M-A1025, respectively. Other loadings of  $\text{MnO}_x$  were also studied however, 20% appears to be optimum.

### 2.2. Catalysts characterization

Specific surface areas were determined using a Quantachrome ChemBET instrument and the single-point Bräuer-Emmett-Teller

(BET) method. All samples were degassed at 130 °C for over 2 h in nitrogen, at a flow rate of 30 ml/min prior to measurements with the nitrogen adsorption isotherm generated at liquid nitrogen temperature 77 K (-196 °C).

X-ray powder diffraction (XRD) patterns were collected on a Rigaku diffractometer using Cu K $\alpha$  radiation ( $\lambda = 0.15406$  nm) at Voltage of 40 kV and current of 200 mA. Intensity data was recorded in  $2\theta$  range of 10–80° with scanning rate of 5°/min.

The thermal gravimetric analysis and differential scanning calorimetry (TG-DSC) profiles were recorded on a simultaneous thermal analyzer (NETZSCH STA 449 F3), with a heating rate of 10 °C/min under a continuous flow of air.

Raman measurements were performed on a Renishaw InVia Reflex micro-Raman spectrometer and recorded at room temperature in an imaging microscope with laser excitation at 514.5 nm in the back-scattering light collection mode. The exciting wavelength of 514.5 nm from an Ar ion laser with a power of 10 mW was used for the catalysts.

Surface Oxidation states were investigated using X-ray photoelectron spectroscopy (XPS), conducted on a PHI5000 Versaprobe-II instruments using a Al-K $\alpha$  monochromatic source. XPS spectra were recorded at room temperature and evacuated to below 10<sup>−7</sup> Pa. The XPS were fitted to investigate the surface active species by measuring the binding energies, the spectral areas and the peak full width at half maxima using Multipak XPS software. All the spectra were calibrated with respect to the binding energy of C 1s signal at 284.80 eV.

Temperature-programmed reduction (TPR) was performed using a Quantachrome instrument with 10%  $\text{H}_2$  in argon at a total flow rate of 25 ml/min. The temperature range studied extended from room temperature to 900 °C at a heating rate of 10 °C/min.

### 2.3. Catalytic activity tests

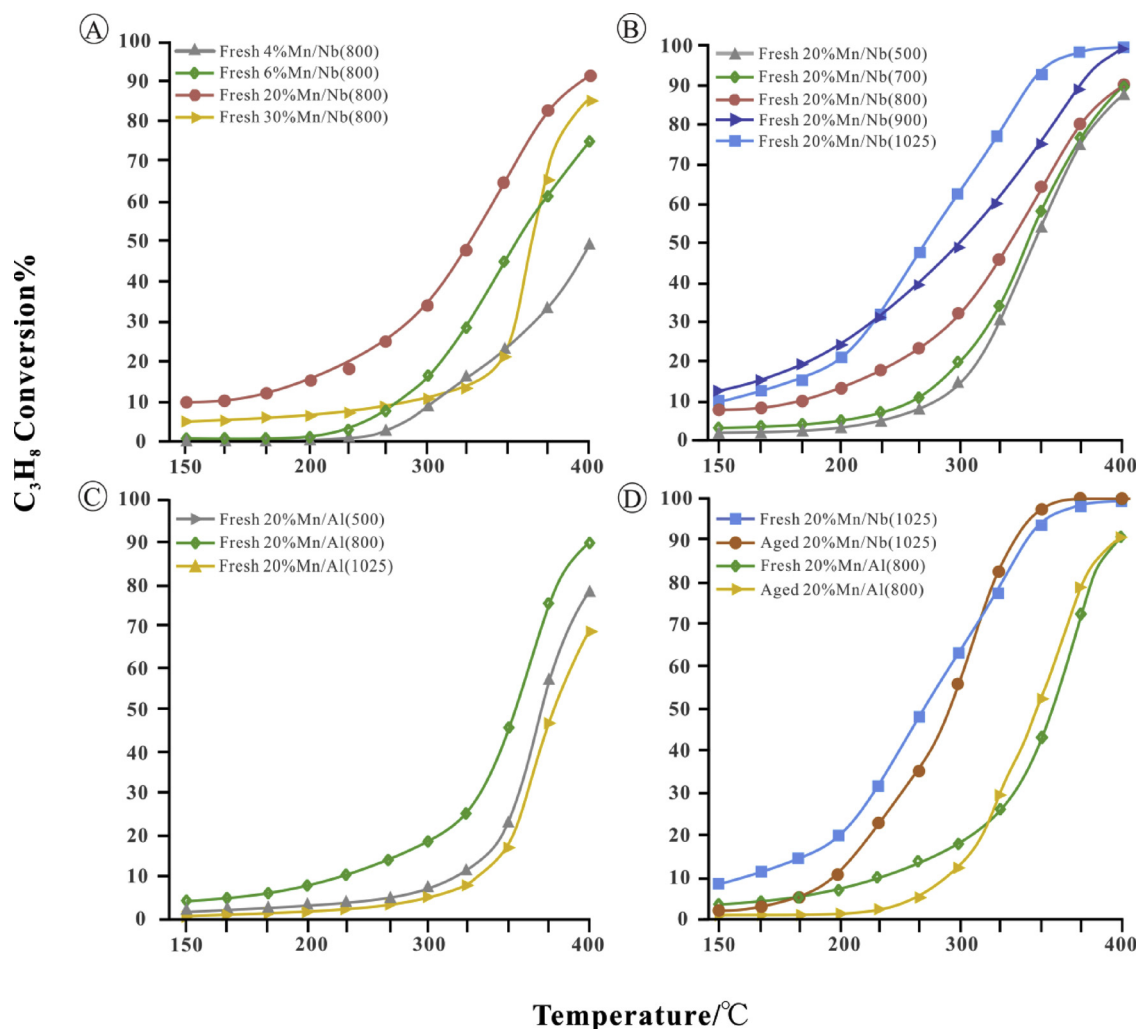
Oxidation tests of propane were carried out in a fixed-bed quartz flow reactor (13 mm diameter) at ambient pressure with the total flow rate of 6.5 L h<sup>−1</sup> and a gas hourly space velocity (GHSV) of 64800 h<sup>−1</sup>. To avoid hot spots from the exothermic reaction and to improve flow distribution, about 0.1 ml of catalyst was diluted with 0.65 g fine quartz packed into the reactor and supported on quartz wool. The catalyst was pre-treated with flowing helium at 400 °C for about 2 h to clean the catalyst surface of volatile species such as absorbed  $\text{H}_2\text{O}$  and  $\text{CO}_2$ .

A cold trap was placed at the outlet of the reactor to condense water produced during the reaction prior to entering the gas chromatography (GC). Effluent gases of  $\text{O}_2$ ,  $\text{CO}_2$ ,  $\text{C}_3\text{H}_8$  and  $\text{N}_2$  were continuously analyzed by the micro GC (INFICON 3000 Micro GC equipped with 10 m Molsieve 5A column, 8 m Plot U column). The GC was calibrated using 4 different certified standard gases before measurements were made. The temperature of the catalyst bed was measured with a K-type thermocouple inside the catalyst bed and controlled by an Omega CN7800 series temperature controller.

For propane oxidation, the final concentration of reactants was 0.1 vol %  $\text{C}_3\text{H}_8$  + 17.85 vol %  $\text{O}_2$  + balance  $\text{N}_2$ . The conversion vs temperature profiles were generated from 25 °C with steady state measurements taken in 25 °C increments up to 425 °C. For all tests, five steady state measurements were made to obtain five data points at each temperature. Carbon balances was calculated for each data point using nitrogen gas as an internal standard. Propane conversions were calculated as follows:

$$\text{Propane conversion (\%)} = \left( \frac{[\text{C}_3\text{H}_8]_{\text{in}} - [\text{C}_3\text{H}_8]_{\text{out}}}{[\text{C}_3\text{H}_8]_{\text{in}}} \right) \times 100$$

Isothermal aging tests were conducted in the process gas at 340 °C for about 30 h (90% conversion) for the best performing sample 20%  $\text{MnO}_x$ / $\text{Nb}_2\text{O}_5$ -x (1025) @400 °C (M-N 1025). After aging the conversion vs temperature profile was again generated from a scan beginning at 25 °C to the final temperature.



**Fig. 1.** Propane conversion versus temperature for various loadings of manganese (A), various pre-treated niobium oxide (B), various pre-treated alumina (C) and comparison of fresh and aged samples of M-N 1025 and M-A800 (D). Aging was conducted at 340 °C for 30 h with reaction gases flowing. Gas composition:  $C_3H_8$  0.1 vol %;  $O_2$  17.85 vol %,  $N_2$  balance. Catalysis volume: 0.1 mL; GHSV: 64,800 ( $h^{-1}$ ).

### 3. Results

#### 3.1. Catalytic performance

The catalytic activity tests for propane were measured from 25 °C to 425 °C, at a GHSV of 64,800 ( $h^{-1}$ ). Various loadings of  $MnO_x$  on  $Nb_2O_{5-x}$  with the latter pre-calcined at various temperatures were studied. The propane conversion vs. temperature plots for a series of  $MnO_x/Nb_2O_{5-x}$  and  $MnO_x/Al_2O_3$  catalysts are presented in Fig. 1, and the reaction temperatures of T20, T50 and T90 (corresponding to propane conversion of 20%, 50% and 90%) are summarized in Table 1 for detailed comparison.

Fig. 1A indicates that Fresh 20%  $MnO_x/Nb_2O_{5-x}$  (800) @400 °C (M-N 800) performs best relative the other loadings. Fig. 1B shows that the highest activity is achieved with 20%  $MnO_x$  deposited on  $Nb_2O_{5-x}$  pre-calcined at 1025 °C (M-N 1025). Fig. 1C demonstrates that  $MnO_x$  over  $\gamma-Al_2O_3$  pre-calcined at 800 °C generates the highest activity. Figs. 1D and 2, however, clearly shows that both fresh and aged 20%  $MnO_x/Nb_2O_{5-x}$  (1025) are more active and stable than 20%  $MnO_x/\gamma-Al_2O_3$  (800 °C) also fresh and aged. Most surprising is the superiority of the 20% Mn catalysts deposited on pre-calcined  $Nb_2O_{5-x}$  (1025) relative to the  $Nb_2O_{5-x}$  calcined at lower temperatures. It was stated earlier that the  $Nb_2O_{5-x}$  (1025 °C) has a SSA < 5  $m^2/g$ , considerably lower than  $Nb_2O_{5-x}$  (> 50  $m^2/g$ ) pre-calcined at lower temperatures. This indicates

**Table 1**

Temperatures (°C) for % conversion of propane.

Catalyst	T20	T50	T90	Simplification
Fresh 4%Mn/Nb(800)@400 °C	335	400	–	
Fresh 6%Mn/Nb(800)@400 °C	310	350	–	
Fresh 30%Mn/Nb(800)@400 °C	355	370	425	
Fresh 20%Mn/Nb(500)@400 °C	295	335	410	M-N500
Fresh 20%Mn/Nb(700)@400 °C	265	330	400	M-N700
Fresh 20%Mn/Nb(800)@400 °C	265	337	395	M-N800
Fresh 20%Mn/Nb(900)@400 °C	198	300	375	M-N900
<b>Fresh 20%Mn/Nb(1025)@400 °C</b>	<b>215</b>	<b>275</b>	<b>340</b>	<b>M-N1025</b>
Fresh 20%Mn/Nb(1025)@500 °C	325	385	–	
Fresh 20%Mn/Nb(1025)@600 °C	365	450	–	
Fresh 20%Mn/Al(500)@400 °C	345	367	–	M-A500
Fresh 20%Mn/Al(800)@400 °C	300	356	400	M-A800
Fresh 20%Mn/Al(1025)@400 °C	350	370	–	M-A1025
<b>Aged 20%Mn/Nb(1025)@400 °C</b>	<b>230</b>	<b>285</b>	<b>335</b>	<b>AGEDM-N1025</b>
Aged 20%Mn/Al(800)@400 °C	314	345	395	AGEDM-A800

a strong positive interaction between  $MnO_x$  and  $Nb_2O_{5-x}$  calcined at 1025 °C, independent of its low surface area. One may postulate that a defect structure has formed leading to enhanced activity, which will be discussed in the following section. Table 1 presents the temperature for 20, 50 and 90% conversion of propane.

To further explore the effect on the activation energy for oxidation

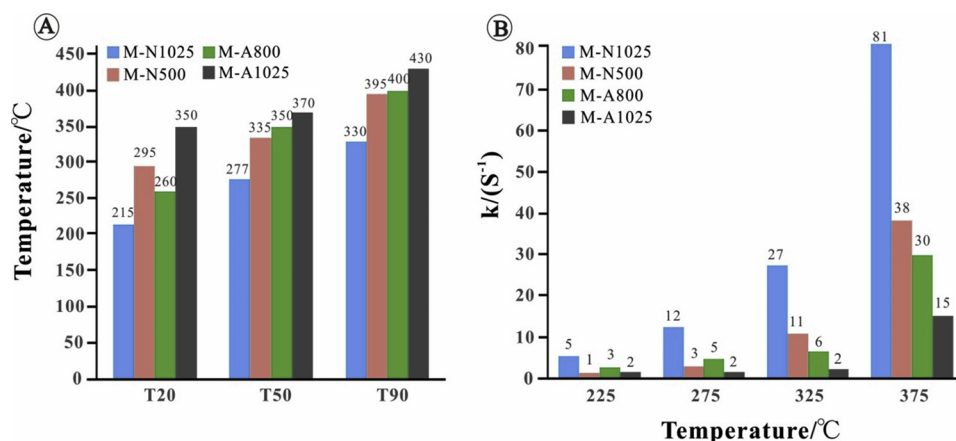


Fig. 2. Comparison of reaction temperature (A) and assumed first order in  $\text{C}_3\text{H}_8$  reaction constants at T20 which is in the kinetic regime.

**Table 2**  
Apparent Activation Energies for the oxidation of propane(kJ/mol).

	M-N1025	M-N500	M-A800	M-A1025
$\text{C}_3\text{H}_8$ Oxidation	39.52	76.04	69.80	121.40

of propane, the kinetic calculations are illustrated below. The assumed first order reaction rate constants of M-N 500, M-N 1025 M-A800 and M-A1025 at different temperatures were calculated (Fig. 2B) while the apparent activation energies ( $E_a$ ) were also obtained (Table 2) based on the catalytic activity data. It should be stressed that, the operation was conducted at low conversions to ensure kinetic control: the GHSV and temperature was adjusted to maintain less than 20%–25% propane conversion. In addition, we also compare the conversion of propane oxidation by diluting 0.1 ml catalysis with quartz from 0.35 g to 0.75 g to ensure isothermal conditions. We have assumed first-order in  $\text{C}_3\text{H}_8$  and pseudo-zero order in oxygen since it is present in a large excess of the pollutants. Rate constants were obtained from equations 1 and 2.

$$\ln\left(\frac{C_i}{C_o}\right) = kt = \frac{k}{\text{GHSV}} \quad (1)$$

$$r = -kC = \left(-A \exp\left(-\frac{E_a}{RT}\right)\right)C \quad (2)$$

where  $C_i$  is the inlet concentration and  $C_o$  is the outlet concentration of reactant and  $A$  is the pre-exponential factor.

The Arrhenius plots for  $\text{C}_3\text{H}_8$  for  $\text{Nb}_2\text{O}_{5-x}$ -based and  $\text{Al}_2\text{O}_3$ -based samples are presented in Fig. 3. The results show that  $\text{Nb}_2\text{O}_{5-x}$ -based samples have relatively lower apparent activation energies and higher first order reaction rate constants compared to  $\text{Al}_2\text{O}_3$ -based samples for propane oxidation. M-N 1025 sample exhibits the highest reaction rate constant ( $k$ ) from 225 °C to 375 °C and lowest apparent activation energy ( $E_a$ ) from T20 to T50 for propane oxidation (39.52 kJ/mol) over the M-N 1025 sample.

### 3.2. Structural properties of catalysts

X-ray powder diffraction was used to study the evolution of the  $\text{Nb}_2\text{O}_{5-x}$  structure with various pre-calcination temperatures and the results are shown in Fig. 4. Patterns for bulk niobium oxide between 500 °C and 1025 °C show various reflections enabling the determination of different phases at each of these temperatures. Three phases of niobium oxide (i.e., pseudo-hexagonal, orthorhombic and monoclinic) are sequentially identified with the temperature increasing from 500 to 1025 °C (Fig. 4A). The pseudo-hexagonal  $\text{Nb}_2\text{O}_{5-x}$  phase is observed when the sample is calcined around 500 °C [31], and the orthorhombic structure is detected when the calcination temperature was increased to

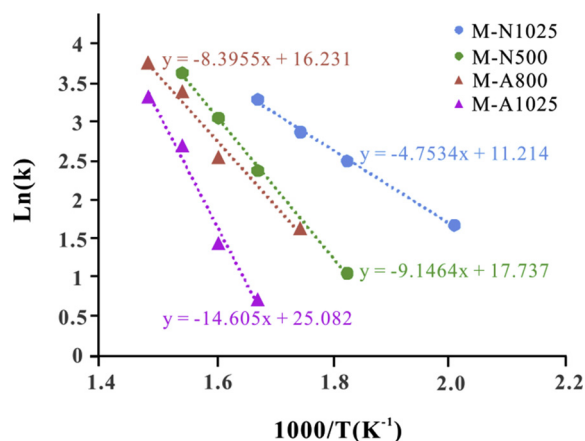


Fig. 3. Arrhenius plots for  $\text{C}_3\text{H}_8$  oxidation under conditions of  $\text{C}_3\text{H}_8$  0.1 vol %;  $\text{O}_2$  17.85 vol %,  $\text{N}_2$  balance with catalysis volume: 0.1 mL and GHSV: 64,800 ( $\text{h}^{-1}$ ).

800 °C [31–35]. Further increasing the temperature to 900 °C very weak peaks corresponding to the monoclinic [33] are identified. For the sample obtained at 1025 °C, both the pseudo-hexagonal and orthorhombic phases change to monoclinic ( $\text{c}2/\text{m}$  and  $\text{P}2$ )  $\text{Nb}_2\text{O}_{5-x}$  [32–36].

Fig. 4B shows the manganese oxide supported on  $\text{Nb}_2\text{O}_{5-x}$  calcined at different temperatures, and the loading content of  $\text{MnO}_x$  is kept at 20 wt% for all samples. All samples show similar XRD patterns with the corresponding pure niobium oxide samples. In addition, there is no obvious shift of patterns for the mixed oxides compared with the pure  $\text{Nb}_2\text{O}_{5-x}$ , indicating that no Mn ions are incorporated into the  $\text{Nb}_2\text{O}_{5-x}$  lattice to form a solid solution. These phenomena are likely to suggest that the size of  $\text{MnO}_x$  and/or other potential Mn-O-Nb compounds are too small to be detected by XRD measurements.

In parallel, The BET results (Supplementary Fig. 1A) show that the surface area of  $\text{Nb}_2\text{O}_{5-x}$  decreases dramatically with increasing pre-calcination temperature due to the formation of new highly structured  $\text{Nb}_2\text{O}_{5-x}$  phases. The surface area of M-N 1025 sample is  $\sim 3.94 \text{ m}^2/\text{g}$  while  $\text{Nb}_2\text{O}_{5-x}$  @1025 is  $2.13 \text{ m}^2/\text{g}$  and  $\text{Nb}_2\text{O}_5$ @500 is about  $50 \text{ m}^2/\text{g}$ . Obviously, these are much lower than  $\gamma\text{-Al}_2\text{O}_3$  and  $\text{Al}_2\text{O}_3$ -based samples ( $100 - 150 \text{ m}^2/\text{g}$ ). Counterintuitively, sample M-N 1025 possesses the lowest surface area but highest catalytic propane activity among all samples evaluated.

Thermogravimetric mass loss (TG) and the associated endotherm of M-N 1025 was determined by the DSC heat flux (mW) between 0–1025 °C. The DSC-TG profile of M-N 1025 sample (Supplementary Fig. 1B) shows an initial endothermic weight loss around 100 °C – 200 °C, corresponding to the loss of water. Low temperature niobium



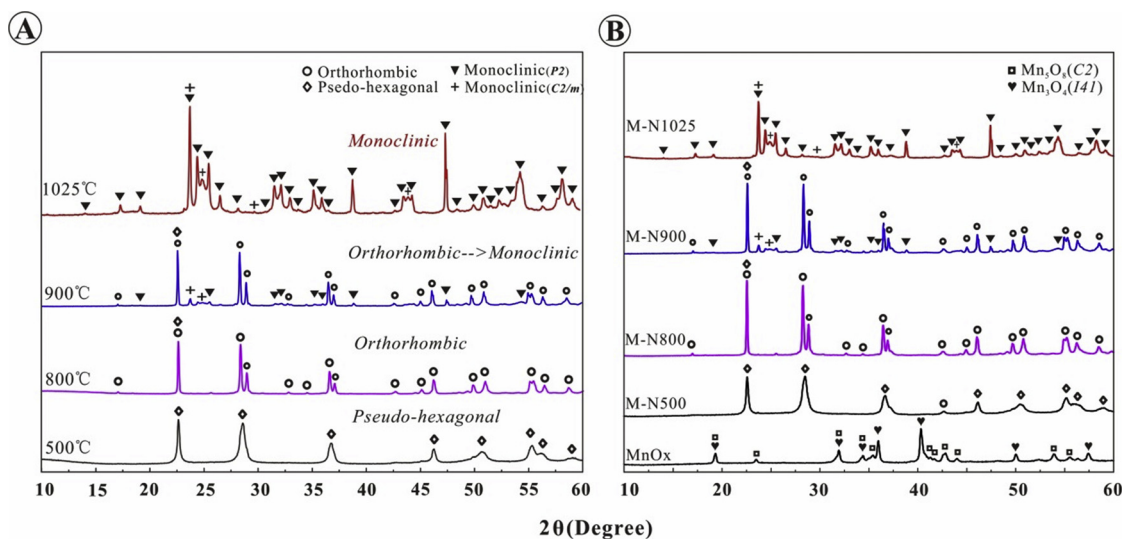


Fig. 4. The X-ray diffraction patterns of aged  $\text{Nb}_2\text{O}_{5-x}$  at different calcined temperatures (A) and 20%MnOx on  $\text{Nb}_2\text{O}_{5-x}$  phases (B).

oxide is stable up to around 300 °C, after which little weight loss occurs with a slowly increasing DSC heat flux signal. A sharp endothermic peak occurs with a small weight loss (~1%) around 570 °C, which could be associated with the phase change of niobium oxide without a change of oxidation state. This seems to be confirmed by XRD pattern above 500 °C.

### 3.3. Surface properties and reducibility

#### 3.3.1. Surface interaction

Raman spectroscopy provides a sensitive response to the detection of the bond order of metal oxides, especially in the region of metal-oxygen stretching modes. A higher metal oxygen bond order, corresponding to a shorter bond distance, shifts the Raman bond to higher wavenumbers [37]. In addition, metal oxides do not always occur in a crystalline form which may not be detected by XRD [38,39].

The Raman spectra of samples of interest are shown in Fig. 5 as a function of pre-calcination temperature. For niobium oxide (Fig. 5A) heated for 3 h at 500 °C, a broad, weak Raman band at ~900  $\text{cm}^{-1}$  and a strong Raman band at ~688  $\text{cm}^{-1}$  is observed. The Raman band of

$\text{Nb}_2\text{O}_{5-x}$  @500 °C in the low-wavenumber range around 240  $\text{cm}^{-1}$  is weak. It should be noticed that the Raman bands below 200  $\text{cm}^{-1}$  are due to lattice vibrations. The Raman features of  $\text{Nb}_2\text{O}_{5-x}$  @800 °C are similar to  $\text{Nb}_2\text{O}_{5-x}$  @500 °C except the intensity at ~688  $\text{cm}^{-1}$  and 200–300  $\text{cm}^{-1}$  is stronger than  $\text{Nb}_2\text{O}_{5-x}$  @500 °C. It is very interesting that, for the  $\text{Nb}_2\text{O}_{5-x}$  sample treated at 900 °C for 3 h, the results indicate that a new phase of  $\text{Nb}_2\text{O}_{5-x}$  sample is initiated and a new small band at ~992  $\text{cm}^{-1}$  is present. Instead of the previous Raman bands at ~688  $\text{cm}^{-1}$ , sample  $\text{Nb}_2\text{O}_{5-x}$  @900 °C possesses twin peaks of the Raman spectra at ~647  $\text{cm}^{-1}$  and ~685  $\text{cm}^{-1}$ . For the  $\text{Nb}_2\text{O}_{5-x}$  @1025 °C sample, the crystal phase transformation is detected by the bands at 992  $\text{cm}^{-1}$  and the twin peak at ~626  $\text{cm}^{-1}$  and ~677  $\text{cm}^{-1}$ , both of them are much sharper and more intense than that of  $\text{Nb}_2\text{O}_{5-x}$  @900 °C. This new phase is also confirmed by XRD and identified as the monoclinic phase (Fig. 4). Furthermore, the bands at ~903  $\text{cm}^{-1}$  and 260  $\text{cm}^{-1}$  (shift from 240  $\text{cm}^{-1}$ ) are more resolved than all samples calcined below 1025 °C. It has been reported that the Raman band at ~930  $\text{cm}^{-1}$  refers to a highly distorted octahedral structure of  $\text{NbO}_6$ , and the strong and twin peaks around ~650  $\text{cm}^{-1}$  assigned to the symmetric stretching mode of the niobia polyhedra [38,39]. The broad

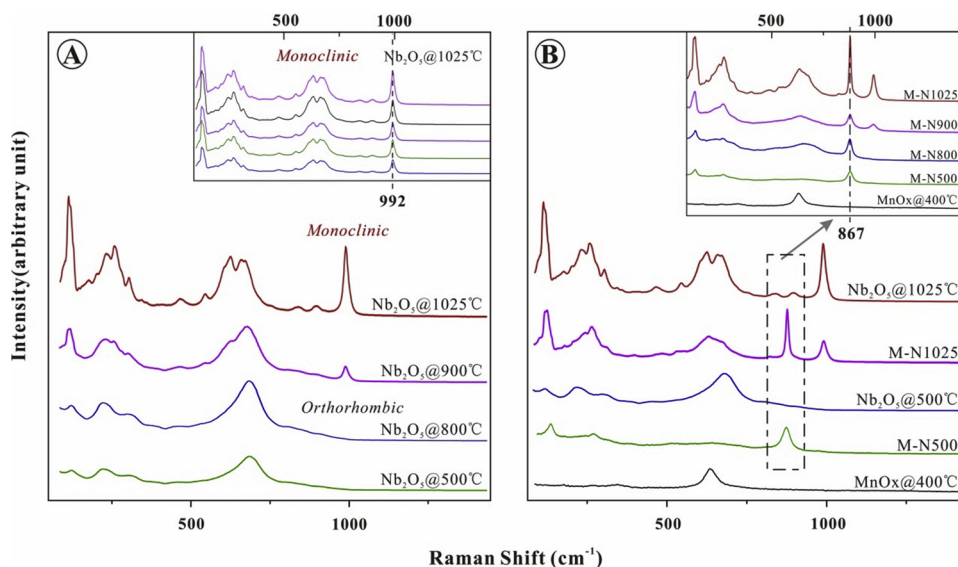


Fig. 5. Raman spectra of bulk  $\text{Nb}_2\text{O}_{5-x}$  at various pre-calcination temperature (A) and (B) Mn-  $\text{Nb}_2\text{O}_{5-x}$ -based catalysts at various temperature.

band at  $\sim 688\text{ cm}^{-1}$  of  $\text{Nb}_2\text{O}_{5-x}$  @  $500^\circ\text{C}$  is characteristic of the vibrations of Nb–O–Nb bridges from slightly distorted octahedral  $\text{NbO}_6$  connected with shared corners. The band between  $150\text{ cm}^{-1}$  and  $300\text{ cm}^{-1}$  is assigned to the bending modes of Nb–O–Nb linkages. The doublet Raman band at  $\sim 626\text{ cm}^{-1}$  and  $\sim 677\text{ cm}^{-1}$  of  $\text{Nb}_2\text{O}_{5-x}$  @  $1025^\circ\text{C}$ , which shifts from  $\sim 688\text{ cm}^{-1}$  for low temperature niobium oxide, associated with vibrations of Nb–O–Nb bridges, suggests that the structures of the niobium oxide species are dependent on the relative pre-calcination temperature [38–40].

Fig. 5B displays the Raman results after 20% $\text{MnO}_x$  was deposited and a sample of pure manganese oxide calcined at  $400^\circ\text{C}$  ( $\text{MnO}_x$ @  $400^\circ\text{C}$ ). All  $\text{Nb}_2\text{O}_{5-x}$ -based catalysts (M-N 500, M-N 800, M-N 900, M-N 1025) show a new band at  $\sim 867\text{ cm}^{-1}$ , which is not observed for pure manganese and/or pure niobium oxide. In contrast, among all  $\text{Al}_2\text{O}_3$ -based samples, only a Raman band of manganese oxide at  $\sim 636\text{ cm}^{-1}$  is observed (Supplementary Fig. 2). Therefore, it is reasonable to deduce that there is an interaction between the oxides of manganese and niobium oxide which may form a unique compound. Secondly, manganese oxide calcined at  $400^\circ\text{C}$  possesses a Raman band at  $\sim 636\text{ cm}^{-1}$ , which is incorporated into the twin peak and then forms a strong and broad Raman spectra  $\sim 636\text{ cm}^{-1}$  with the M-N 1025 sample, whereas in M-N 500, the counterpart of Raman band is not observed. Thirdly, Raman studies reveal that the new compound containing oxides of manganese and niobium possesses a similar structure to that of  $\text{H-Nb}_2\text{O}_5$  ( $\text{Nb}_2\text{O}_{5-x}$  @  $1025^\circ\text{C}$ ) (Fig. 5B) since apart from the new peaks ( $\sim 867\text{ cm}^{-1}$ ) of the spectra, M-N 1025 maintains the characteristic bands of  $\text{Nb}_2\text{O}_{5-x}$  @  $1025^\circ\text{C}$  at  $\sim 992\text{ cm}^{-1}$  and  $260\text{ cm}^{-1}$  without shifting.

### 3.3.2. Surface oxidation states and defect sites

To further confirm the surface composition and chemical state of the atoms in  $\text{Nb}_2\text{O}_{5-x}$ -based samples, X-ray photoelectron spectroscopy (XPS) was conducted with results shown in Fig. 6. The reference binding energy for  $\text{Nb}^{5+} 3d_{5/2}$ ,  $\text{Nb}^{4+} 3d_{5/2}$  and  $\text{Nb}^{2+} 3d_{5/2}$  core levels are listed in the Supplementary Table 1 [41–46]. The quantitative analysis of relative content of different ions (e.g.,  $\text{Nb}^{5+}$ ,  $\text{Nb}^{4+}$ ,  $\text{Mn}^{4+}$  and  $\text{Mn}^{3+}$ ) on the surface is obtained by calculating the area ratios of corresponding species derived from the raw XPS data, as shown in Table 3.

Fig. 6A shows the evolution of the Nb3d doublet with increasing pre-calcination temperature of the  $\text{Nb}_2\text{O}_{5-x}$  only. The doublet seen in the Nb 3d spectra constitutes peaks of  $\text{Nb} 3d_{5/2}$  and  $\text{Nb} 3d_{3/2}$  with the oxide states of Nb analyzed by fitting the curves of the Nb  $3d_{5/2}$  XPS bands. The entire position of Nb  $3d_{5/2}$  deconvoluted experimental XPS bands (Fig. 6A) indicate  $\text{Nb}^{5+}$  (207.17 eV–207.52 eV) and  $\text{Nb}^{4+}$  (205.38 eV–206.81 eV) oxidation state, which consistent with the binding energies in the literature [41–48]. More specifically, the Nb 3d doublet of the original data of XPS occur at lower binding energies with increasing pre-calcination temperature of the niobium oxide (from M-N 500 to M-N 1025). This is likely due to a change in chemical environment of the niobium atoms by lattice deformation induced a stronger deformation niobium oxide when calcined at higher temperature. Accordingly, it is found that with the loading of 20%  $\text{MnO}_x$ , the average ratios of  $\text{Nb}^{4+}/\text{Nb}^{5+}$  increases in the order of M-N 500 < M-N 800 < M-N 900 < M-N 1025 (Table 3). The highest ratio of  $\text{Nb}^{4+}/\text{Nb}^{5+}$  in this study is 0.92 for sample M-N 1025. In other words, the  $\text{Nb}_2\text{O}_{5-x}$  pre-calcined at  $1025^\circ\text{C}$  has a composition of  $\text{Nb}_2\text{O}_{5-x}$  where  $0 < x < 1$ , which indicates oxygen loss from the surface. In addition, the experimental Nb  $3d_{5/2}$  could also be deconvoluted into the two contributions of  $\text{Nb}^{4+}$  and  $\text{Nb}^{5+}$  with a  $\text{Nb}^{4+}/\text{Nb}^{5+}$  ratio of 0.51 in the sample of pure niobium oxide calcined at  $1025^\circ\text{C}$ . The peak of  $\text{Nb}^{5+}$  is the only component analyzed by XPS in the sample of pure  $\text{Nb}_2\text{O}_{5-x}$  @  $500^\circ\text{C}$ . The highest ratio of  $\text{Nb}^{4+}/\text{Nb}^{5+}$  achieved in the niobium pre-calcined at  $1025^\circ\text{C}$  indicates that the pre-calcination temperature of niobium oxide plays an important role in the oxidation state and thus catalytic activity.

The O 1s spectra of M-N 500, M-N 800, M-N 900 and M-N 1025 are

shown in Fig. 6B depends on both the chemisorbed oxygen species ( $\text{O}_\alpha$ ) and the lattice oxygen ( $\text{O}_\beta$ ). It should be noted that the peak for adsorbed oxygen is generally weaker than that of lattice oxygen and could be an indicator of oxygen vacancies [47–49]. As a whole, the peak at 529.40–529.84 eV corresponds to  $\text{O}_\beta$ , whereas the experimental bands at 530.89–532.00 eV can be ascribed to surface-adsorbed oxygen species ( $\text{O}_\alpha$ , i.e.  $\text{O}_2^{2-}$ ,  $\text{O}^-$ ,  $\text{OH}^-$  or  $\text{CO}_3^{2-}$ ) [16,50,51]. The O 1s spectra could be deconvoluted into two contributions, surface oxygen and lattice oxygen, whereas the M-N 1025 sample could be fitted with three peaks at 532.00 eV, 530.89 eV and 529.40 eV. The former two could be assigned to surface oxygen with various adsorbed oxygen species, and the latter attributed to the lattice oxygen. It is generally accepted that the surface adsorbed oxygen ( $\text{O}_\alpha$ ) possesses greater mobility than lattice oxygen and may give rise to beneficial spillover phenomena at the solid surface [16,19]. As illustrated in Table 3, the highest  $\text{O}_\alpha/\text{O}_\beta$  ratio is attained for sample M-N 1025 ( $\text{O}_\alpha/\text{O}_\beta = 1.47$ ), confirming the high proportion of oxygen vacancies present. In associated with the XPS data, obtained for pure niobium oxide calcined at different temperatures, shows an increasing  $\text{O}_\alpha/\text{O}_\beta$  ratio as follows: M-N 500  $\approx$   $\text{Nb}_2\text{O}_{5-x}$  @ 500 < M-N 800 < M-N 900 <  $\text{Nb}_2\text{O}_{5-x}$  @ 1025 < M-N 1025 (Table 3), which again indicates the effect of pre-calcined niobium on the presence defect structures. This result is consistent with Nb 3d.

Fig. 6C shows the Mn 2p XPS spectra of samples in a range between 635 eV and 662 eV. The binding energies at 640.00–650.00 eV and 650.00–660.00 eV can be attributed to Mn  $2p_{3/2}$  and Mn  $2p_{1/2}$ , respectively [43,52]. The deconvoluted Mn 2p spectra for all samples could be assigned to three contributions, the lowest binding energy being ascribed to  $\text{Mn}^{2+}$  at  $\sim 640.98\text{ eV}$ – $641.68\text{ eV}$ . The range of 642.06 eV to 643.19 eV could be assigned to  $\text{Mn}^{3+}$ .  $\text{Mn}^{4+}$  has the highest binding energy and can be assigned range of 644.64–645.02 eV. These results are in agreement with the literature [43,52–55]. Table 3 lists the relative surface composition of manganese with different valences. The highest  $\text{Mn}^{3+}/\text{Mn}^{2+}$  and  $\text{Mn}^{3+}/\text{Mn}^T$  ratio obtained is from M-N 1025, whereas the lowest ratios occur for M-N 500 (Table 3). It is possible that the increasing component of the  $\text{Mn}^{3+}$  &  $\text{Mn}^{4+}$  pair is associated with the active catalytic states. It is possible that the niobium oxide calcined above  $1000^\circ\text{C}$  has high interaction with manganese oxide and consequently induces an increasing proportion of active phases of manganese on the surface of catalyst. This tentative conclusion is in fair agreement with the Raman band at  $867\text{ cm}^{-1}$  (Fig. 5).

### 3.3.3. Reducibility

$\text{H}_2$ -TPR was conducted to test the reducibility of samples. According to the literature [56,57], the reduction of manganese oxides can be described by the following steps:  $\text{MnO}_2 \rightarrow \text{Mn}_2\text{O}_3 \rightarrow \text{Mn}_3\text{O}_4 \rightarrow \text{MnO}$ . However, the TPR profiles could also depend on the Mn cations located in different environments [58].

Fig. 7 shows the  $\text{H}_2$ -TPR profiles of the fresh manganese oxide catalysts supported on different  $\text{Nb}_2\text{O}_{5-x}$  and  $\text{Al}_2\text{O}_3$  carriers as the function of reduction temperature ( $^\circ\text{C}$ ). In M-N 1025 the most active catalyst, three peaks are observed whereas only two peaks are observed in the other samples. The two peaks at  $282^\circ\text{C}$  and  $307^\circ\text{C}$  could be ascribed to the reduction of  $\text{MnO}_2$  ( $\text{MnO}_2$  to  $\text{Mn}_2\text{O}_3$ ) and the reduction of  $\text{Mn}_2\text{O}_3$  ( $\text{Mn}_2\text{O}_3$  to  $\text{Mn}_3\text{O}_4$ ), respectively. The last broad peak at  $385^\circ\text{C}$  could be attributed to the reduction of  $\text{Mn}_3\text{O}_4$  ( $\text{Mn}_3\text{O}_4$  to  $\text{MnO}$ ). In contrast only two peaks are observed in the TPR profile for all the other samples. In the case of M-N 900, M-N 800, M-N 500 and M-A800, the two peaks could be attributed to the reduction of  $\text{Mn}_2\text{O}_3$  ( $\text{Mn}_2\text{O}_3$  to  $\text{Mn}_3\text{O}_4$ ) and  $\text{Mn}_3\text{O}_4$  ( $\text{Mn}_3\text{O}_4$  to  $\text{MnO}$ ), respectively. The lowest reduction temperature ( $305^\circ\text{C}$ ) for  $\text{Mn}_2\text{O}_3$  was obtained by M-N 900, whereas  $345^\circ\text{C}$  was obtained for M-N 800. Furthermore, the reduction temperature of  $\text{Mn}_2\text{O}_3$  follows the order of M-N 1025 < M-N 900 < M-N 800 < M-N 500  $\approx$  M-A800. It is reasonable to conclude that the M-N 1025 sample has the highest reducibility which generates higher catalytic oxidation activity when interacting with the manganese species. The reduction of  $\text{MnO}_2$  is only observed in M-N 1025 at  $282^\circ\text{C}$  which

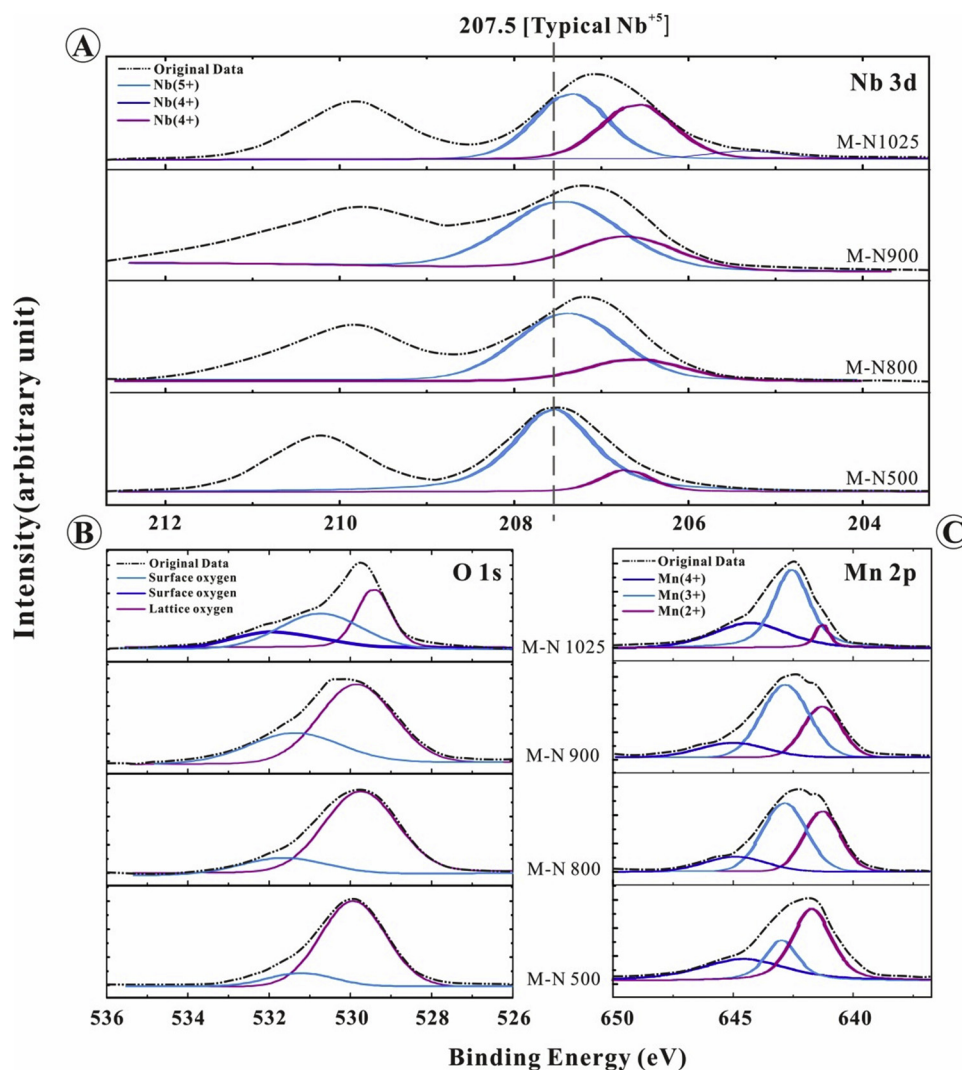


Fig. 6. Nb 3d (A), O 1s (B) and Mn 2p<sub>3/2</sub>(C) XPS spectra of Nb<sub>2</sub>O<sub>5-x</sub>-based samples.

likely correlates with the relative larger amounts of Mn<sup>4+</sup> on the surface. Negligible amounts of Mn<sup>4+</sup> species are present on the surface of low temperature calcined niobium oxide and  $\gamma$ -alumina. Another possibility for the peak of MnO<sub>2</sub> reduction could be the synergistic effect between Mn and Nb on sample M-N 1025, which enhances the redox properties of Mn<sub>4</sub><sup>+</sup>/Mn<sub>3</sub><sup>+</sup> pair while not for the others.

#### 4. Discussion

For propane oxidation on niobium-based catalysts, a number of parameters, such as catalyst crystallite size, surface area, reducibility of

the manganese species, and manganese content are believed to impact catalytic performance [57–60]. Generally, BET surface area and the concentration of manganese are usually important factors in determining catalytic activity. This is clearly not the case of for niobium oxide as a carrier material for MnO<sub>x</sub>. Certainly our data indicates that the catalytic activity of MnO<sub>x</sub>/Nb<sub>2</sub>O<sub>5-x</sub> increases with manganese loading up to 20%, but performance declines at a 30% loading (Fig. 1A). The data for MnO<sub>x</sub> on Nb<sub>2</sub>O<sub>5-x</sub> (1025 °C) is consistent with our previous observations on Co/NbO<sub>x</sub> which also shows an increased activity for pre-calcined Nb<sub>2</sub>O<sub>5-x</sub> [30] independent of BET surface area. For example, the surface area increases in the order of M-N 1025

**Table 3**  
XPS results of the Mn2p<sub>3/2</sub>, Nb3d<sub>5/2</sub> and O 1s for Nb<sub>2</sub>O<sub>5-x</sub>-based catalysts.

Catalysts	Relative Surface Content						Surface Atom Ratio		
	Mn 2p <sub>3/2</sub>			Nb 3d <sub>5/2</sub>			O 1s		
	Mn <sup>3+</sup> /Mn <sup>2+</sup>	Mn <sup>3+</sup> /Mn <sup>4+</sup>	Mn <sup>3+</sup> /Mn <sup>T</sup>	Nb <sup>4+</sup> /Nb <sup>5+</sup>	Nb <sup>4+</sup> /Nb <sup>T</sup>		O <sub>α</sub> (%)	O <sub>β</sub> (%)	O <sub>α</sub> /O <sub>β</sub>
M-N1025	9.29	1.89	0.61	0.92	0.48		59.39	43.41	1.47
M-N900	1.8	3.58	0.54	0.41	0.29		29.7	70.3	0.42
M-N800	1.36	2.83	0.48	0.15	0.13		19.5	80.5	0.24
M-N500	0.51	0.70	0.23	0.13	0.11		12.88	87.12	0.15
									Mn/Nb/O
									28/4/68
									16/13/71
									16/14/70
									10/20/70

<sup>T</sup>Mn<sup>T</sup> = (Mn<sup>2+</sup> + Mn<sup>3+</sup> + Mn<sup>4+</sup>); Nb<sup>T</sup> = (Nb<sup>4+</sup> + Nb<sup>5+</sup>).



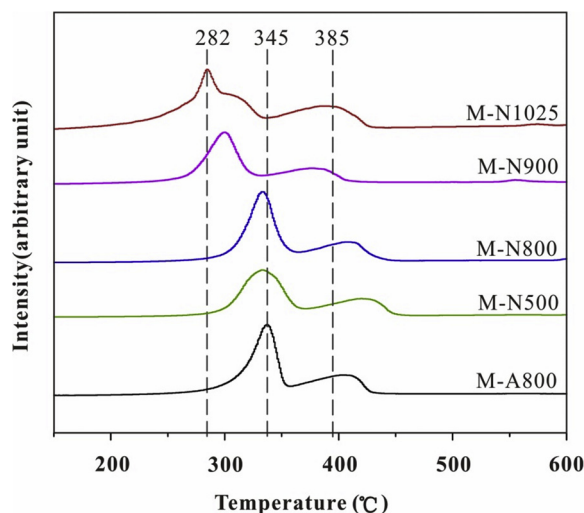


Fig. 7.  $\text{H}_2$ -TPR profiles of the  $\text{MnO}_x$ - $\text{Nb}_2\text{O}_{5-x}$ -based and  $\text{Al}_2\text{O}_3$ -based samples.

( $2.13 \text{ m}^2/\text{g}$ ) < M-N 900 < M-N 800 < M-N 500 ( $\sim 50 \text{ m}^2/\text{g}$ ), whereas the catalytic performance shows the opposite trend. Although the catalytic performance increases with increasing the pre-calcination temperatures of  $\text{Nb}_2\text{O}_5$ , the most dramatic increase occurs above  $1000^\circ\text{C}$  which is related to a phase transition to monoclinic. The catalytic performance coupled with characterization data lead us to conclude the phase change effect of  $\text{Nb}_2\text{O}_{5-x}$  @ $1025^\circ\text{C}$  on improving the catalytic properties with the following two possible reasons: (1) the presence of oxygen vacancies and (2) the proper  $\text{Mn}^{3+}/\text{Mn}^{4+}$  ratios induced by the interaction between  $\text{Nb}_2\text{O}_{5-x}$  @ $1025^\circ\text{C}$  and  $\text{MnO}_x$ .

#### 4.1. Effect of pre-calcination temperature on the structure of $\text{Nb}_2\text{O}_5$

The increasing pre-calcination temperature of niobium oxide leads to phase transformations and a mixture of monoclinic phases of  $\text{Nb}_2\text{O}_{5-x}$  containing non-stoichiometric oxides ( $\text{Nb}_{12}\text{O}_{29}$ ) formed around  $1025^\circ\text{C}$ . This generates an abundance of oxygen vacancies, beneficial to propane oxidation.

Consistent with characterization results in the literatures [31–35,61], it is reasonable to propose the structure evolution of niobium oxide with increasing calcination temperature follows the schematic diagram Fig. 8. Below or around  $500^\circ\text{C}$ , niobium oxide remains amorphous but then changes to a pseudo-hexagonal structure (TT-

$\text{Nb}_2\text{O}_5$ ) [31,35]. Orthorhombic crystal (T- $\text{Nb}_2\text{O}_5$ ) could be obtained at temperatures of  $600 - 800^\circ\text{C}$  [31,34–36]. One of the monoclinic phases initially formed above  $900^\circ\text{C}$  and then achieves the mixture of monoclinic  $\text{Nb}_2\text{O}_5$  and monoclinic  $\text{Nb}_{12}\text{O}_{29}$  around  $1025^\circ\text{C}$  where stoichiometric  $\text{Nb}_2\text{O}_5$  converts to a higher proportion of non-stoichiometric niobium oxide, generating oxygen vacant sites [32].

The XRD results (Fig. 4) of M-N 1025 sample and  $\text{Nb}_2\text{O}_{5-x}$  @ $1025^\circ\text{C}$  sample exhibit a mixture of monoclinic phases with spatial group of  $\text{P}_2$  and  $\text{C}_2/\text{m}$ , which contain non-stoichiometric monoclinic niobium oxides,  $\text{Nb}_{12}\text{O}_{29}$ . This is consistent with the Raman structure (Fig. 5) of M-N 1025 which possesses the highly distorted cell of  $\text{NbO}_6$  octahedra, where the cells are linked by edge-sharing. The non-stoichiometric niobium oxides formed due to the crystallographic shearing of highly distorted octahedral cells of  $\text{NbO}_6$  unit above  $1000^\circ\text{C}$  where the niobium oxide octahedral changes its linkage from corner to edge-sharing and possess the Nb atoms displaced from the center of the  $\text{NbO}_6$  cell [32,62–64]. This kind of shearing plane and the modified linkage type of  $\text{NbO}_6$  blocks facilitate the formation of monoclinic niobium with non-stoichiometric component, which in turn allow the niobium ion coordination to accommodate large oxygen deficiencies [62,64]. Marucco [65] previously observed  $\text{Nb}_{12}\text{O}_{29}$  stable at temperature between  $1000^\circ\text{C}$  and  $1100^\circ\text{C}$ , later, Kikuchi et al. [66] and Iijima et al. [67,68] confirmed oxygen vacancies using high resolution electronic microscopy which displayed complex point defects in f  $\text{Nb}_{12}\text{O}_{29}$ . In addition, Janninck et al. [69] and Marucco [70] reported the oxygen vacancies in monoclinic niobium oxide with stoichiometry deviations by measuring the electrical conductivity as a function of oxygen partial pressure.

#### 4.2. Effect of $\text{Nb}_2\text{O}_{5-x}$ structure on the interaction between niobium and manganese

A set of  $\text{MnO}_x/\text{Nb}_2\text{O}_{5-x}$  catalysts were prepared by an incipient wetness impregnation and with niobium oxide pre-calcined from  $500^\circ\text{C}$  to  $1100^\circ\text{C}$ . The catalytic properties were evaluated by the total oxidation of propane. It was reported that hydrocarbon oxidation (i.e. propane) over manganese oxides occurs through a Mars-Van Krevelen (M-K) mechanism involving the participation of lattice oxygen by a redox cycle [58–60,71,72]. The presence of higher surface oxygen species and oxygen vacancies in sample M-N 1025 provides relatively high freedom for the movement of lattice oxygen. Owing to the enrichment of oxygen vacancies, the absorbed oxygen could reduce the surface Gibbs energy and form surface active oxygen species, which could ultimately promote oxygen reactivity and propane activation in oxidation reactions [73]. In association with the effect of abundant

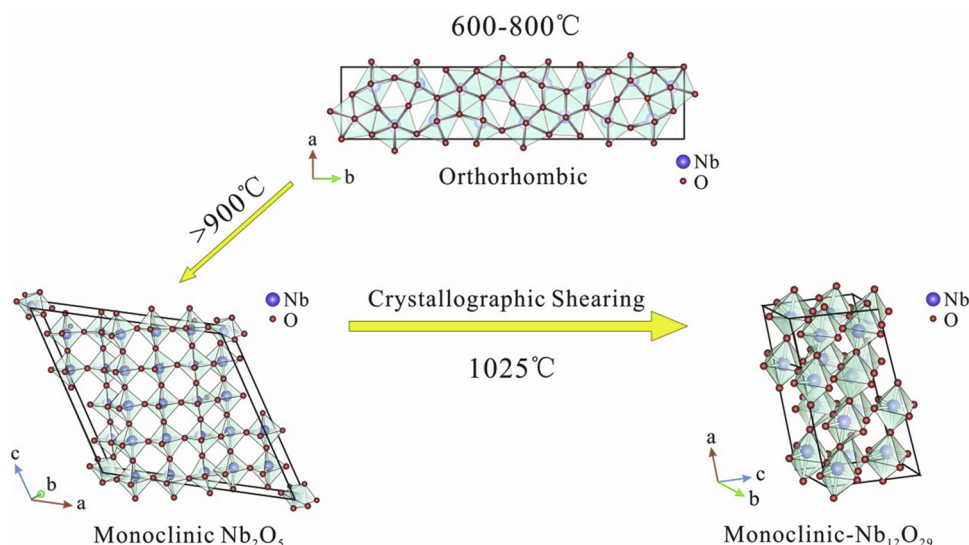


Fig. 8. Structure evolution of niobium oxide calcinated at different temperatures (Modified after [31–35]).



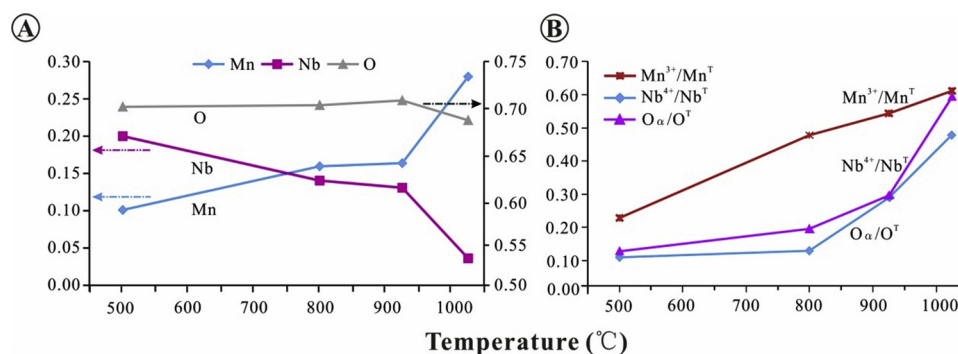


Fig. 9. Relative ratio of surface atom (A) and surface valence content (B) as a function of MnO<sub>x</sub> supported by Nb<sub>2</sub>O<sub>5-x</sub> obtained at different temperatures.

oxygen vacancies, the enhanced redox properties of the surface Mn<sup>3+</sup>/Mn<sup>4+</sup> pair raised by the interaction between manganese and active niobium oxide contribute to the decreasing of apparent activation energy for this reaction. In fair agreement with this mechanism, a relatively low apparent activation energy for propane oxidation obtained in M-N 1025 (39.52 kJ/mol) of this study.

The surface composition and valence states of metal ions were further analyzed. Based on the aforementioned results and the literature, a stronger interaction between Nb<sub>2</sub>O<sub>5-x</sub> @1025 °C and MnO<sub>x</sub> leads to high surface Mn<sup>3+</sup> and the proper ratio of Mn<sup>3+</sup>&Mn<sup>4+</sup> which are the active promoters for the total oxidation of propane.

The Raman results demonstrated that (Fig. 6) as the pre-calcination temperature of niobium species increases, the new peak of Nb<sub>2</sub>O<sub>5-x</sub> based samples become sharper and more intense at 867 cm<sup>-1</sup> and the twin peak of manganese oxide at ~636 cm<sup>-1</sup> becomes much stronger and broader accordingly. The new peak around 867 cm<sup>-1</sup> observed in Raman spectra associated with XPS analysis (Fig. 9) indicate the corresponding dependence of the pre-calcination temperature on the surface concentration of manganese, niobium and oxygen. As the pre-calcination temperature of niobium oxide increases from 500 °C to 1025 °C, the relative content of surface manganese increases from 10% to 28% while niobium decreases from 20% to 4% (Fig. 9A). This phenomenon could be ascribed to the interaction of manganese species with Nb<sub>2</sub>O<sub>5-x</sub>@1025, in which the unique structure of higher temperature niobium oxide is capable to induce enrichment of surface-active manganese species. Fig. 9B exhibits that, among all species on the catalyst surface, the remarkably increasing ratio of Mn<sup>3+</sup>/(Mn<sup>2+</sup> + Mn<sup>3+</sup> + Mn<sup>4+</sup>) occurs with increasing pre-calcination temperature revealing the distinct effect of niobium oxide phases on the surface valence states of the manganese species. It should be stressed that the Nb<sup>4+</sup>/Nb<sup>5+</sup> ratio increases with the increase of Nb<sup>4+</sup> due not only to the phase change of niobium oxide itself, but also the interaction with the surface manganese species. Additionally, the comparably high content of Mn<sup>4+</sup> present on Nb<sub>2</sub>O<sub>5-x</sub> @1025 °C is further substantiated by H<sub>2</sub>-TPR profile (Fig. 7), in which the reduction of MnO<sub>2</sub> (MnO<sub>2</sub> to Mn<sub>2</sub>O<sub>3</sub>) exclusively observed in M-N 1025. Apart from the strong interaction between high temperature niobium oxide and manganese oxide, those results demonstrate the active catalytic component is the Mn<sup>3+</sup>&Mn<sup>4+</sup> couple as well. In addition to the proper content of Mn<sup>3+</sup>&Mn<sup>4+</sup>, the interaction between manganese and niobium species are capable to further enhance the oxygen vacancy of the material [74]. The increased oxygen vacancies formed in the Mn-O-Nb material in turn increase the mobility of oxygen in the lattice of manganese, which ultimately enhances the catalytic oxidation properties. It is fair agreement with the literature that Mn<sub>3</sub>O<sub>4</sub> with the surface partially reduced species possesses a high concentration of oxygen vacancies and higher catalytic activity for hydrocarbon oxidation rather than a surface with a similar stoichiometry manganese oxide [17,18].

The generally consistent results obtained from Raman, XPS and TPR in association with the literature demonstrate the stronger interaction

occurs between Nb<sub>2</sub>O<sub>5-x</sub> @1025 °C and MnO<sub>x</sub>, providing the proper ratio of Mn<sup>3+</sup> & Mn<sup>4+</sup> and high oxygen mobility on the catalyst surface. Therefore, it is reasonable to propose that a new Mn-O-Nb specie with a large amount of active manganese is formed in the M-N 1025 sample, which could largely enhance the catalytic activity for propane oxidation by achieving the synergistic catalytic effect between MnO<sub>x</sub> and Nb<sub>2</sub>O<sub>5-x</sub> @1025 °C.

## 5. Conclusion

Manganese oxides supported on niobium oxide have been extensively characterized and evaluated for total oxidation of propane. Niobium oxide pre-calcined above 1000 °C forms a specific monoclinic phase giving the sample M-N 1025 high performance for propane oxidation, regardless of very low specific BET surface area (3.94 m<sup>2</sup>/g). The distinctive catalytic performance is further shown with an activation energy of 39.52 kJ/mol for propane combustion, which is much lower than that for the M-N 500 sample (76.04 kJ/mol for propane combustion).

XRD, Raman and XPS have been used to identify the presence of an abundance of oxygen vacancies in monoclinic- Nb<sub>2</sub>O<sub>5-x</sub> with a high ratio of Nb<sup>4+</sup>/Nb<sup>5+</sup> coupled with the Mn<sup>3+</sup>/Mn<sup>T</sup> ratio induced by the strong interaction between the two. These results agree with the enhanced performance of C<sub>3</sub>H<sub>8</sub> combustion. A high ratio of Nb<sup>4+</sup>/Nb<sup>5+</sup> appears to be the most significant surface composition in its promoting effect on MnO<sub>x</sub>. The excellent catalytic performance reveals that the phase transformation of niobium oxide above 1000 °C is one of the most important factors that influences the catalytic activity. Additionally, the distinctive catalyst-carrier interaction between manganese and monoclinic- Nb<sub>2</sub>O<sub>5-x</sub> achieves and enhances the synergistic catalytic effect.

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## Appendix A. Supplementary data

Supplementary material related to this article can be found, in the online version, at doi:<https://doi.org/10.1016/j.apcatb.2018.11.066>.

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